Post-functionalization of graphitic carbon nitride for highly efficient photocatalytic hydrogen evolution

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Abstract: In this work we report the feasible modification of graphitic carbon nitride $(g-C_3N_4)$ polymer through a postfunctionalization progress. The resultant photocatalyst exhibits boron doping and mesoporous structure with a high surface area of 125 m²/g, leading in an increased surface activity for photocatalytic water splitting reaction. X-ray diffraction, X-ray photoelectron spectroscopy, PL emission spectra and UV-Vis spectra were used to detect the properties of as-prepared samples. Based on X-ray photoelectron spectroscopy analysis, boron is proposed to dope in the g-C₃N₄ lattice. Optical studies indicated that boron doped g-C₃N₄ exhibits enhanced and extended light absorbance in the visible-light region and a much lower intensity of PL emission spectra compared to pure g-C₃N₄. As a result, boron doped g-C₃N₄ shows activity of 10.2 times higher than the pristine g-C₃N₄ for photocatalytic hydrogen evolution. This work may provide a way to design efficient and mesoporous photocatalysts through post modification.

Key words:boron-doping; mesoporous structure; post-functionalization; photocatalytic hydrogen evolutionCLC number:0644Document code:A

The awareness of energy conservation and environmental protection has been roused due to the worldwide energy shortage and environmental issues, while numerous research focus on the exploration of renewable and clean energy^[1]. Since the first discovery photoelectrochemical (PEC) water of splitting processing with titanium dioxide $(TiO_2),$ semiconductor-based photocatalysis for solar hydrogen production has attracted global interests^[2-4]. However, it is still challenging to achieve high solar-to-hydrogen (STH) conversion efficiency due to the restricted utilization of sunlight. To make good use of solar energy, the development of visible light responsive photocatalysts is highly desirable. In 2009, Wang et al.^[5,6] developed graphitic carbon nitride $(g-C_3N_4)$ for visible-light-driven photocatalytic water splitting, making it a hot research topic in the last decade.

Unlike TiO₂, $g-C_3N_4$ affords a narrow bandgap of ca. 2.7 eV with the conduction band (CB) position at ca. -1.1 eV and valance band (VB) position at ca. +1.6 eV vs. normal hydrogen electrode (NHE), respectively^[5]. Furthermore, $g-C_3N_4$ photocatalyst

possesses the advantages of stability, non-toxicity, abundant source, absorption in visible-light region and easy to control and modify. As a novel visible-lightdriven photocatalyst, its remarkable property has been demonstrated in various photocatalytic applications such as H₂ and O₂ evolution^[7-10], photodegradation of pollutants^[11], CO₂ reduction^[12], organic synthesis^[13] and photoelectrocatalysis^[14]. However, the three pivotal issues of light absorption efficiency, charge separation efficiency and surface reaction efficiency still limit the performance of g-C₃N₄-based photocatalysis. Therefore, it is highly desirable to modify g-C₃N₄ for efficient solar water splitting. Doping is considered as a very convenient and promising strategy to tune the band structure, enhance the lighting adsorption and improve the performance of photocatalyst^[15]. Various doping strategies are employed to modify the g-C₃N₄ to improve the light harvesting, suppress the photogenerated charge recombination and enhance the photocatalytic activity. Combining g-C₃N₄ with strongly electronegative dopants such as fluorine to form an F-doped material not only raised the valence

Received: 2022-04-06; Revised: 2022-05-06

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The project was supported by Natural Science Youth Foundation of Henan Province (202300410032), Key Scientific Research Projects of Colleges and Universities of Henan Provincial Department of Education (21A150010), Foundation for University Key Teacher by the Henan University of Urban Construction (YCJQNGGJS202109).

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band, but also effected the thermodynamic driving force for H₂ reduction^[16,17]. Fluorinated g-C₃N₄ solids also reported with excellent visible-light was photocatalytic activity^[18]. Besides fluorine, I doping also lead positive effect like enlarged specific surface area, enhanced optical absorption, narrowed bandgap and accelerated charge carriers transfer rate as well as increased H₂ evolution rate^[19]. Moreover, potassiummodified $g-C_3N_4$ (K-g-C₃N₄) nanosheets were synthesized^[20]. Photocatalytic H₂ evolution experiments under visible light irradiation showed that K-g-C₃N₄ nanosheets have high photocatalytic activities (up to about thirteen times higher than that of pure $g-C_3N_4$) as well as good stability (no reduction in activity within 16 h). However, the doping methods have been reported mainly via calcination of precursors and doping source, which results in the bulk structure and heterogeneous distribution of doped ions and limited enhanced activity of photocatalysis. Therefore, it is of great importance to develop a feasible one-pot route for forming the porous structure and homogenizing the doping ions, which induces more reaction sites and optimizes the electronic structure. Furthermore, the selection of boron source may show significant effect on the doping sites and Lewis acidity on the surface of g-C₃N₄, which may also act as specific reaction sites for reactant molecules^[21]. In particular, the textural, electronic, and structural properties of g-C₃N₄ can be modulated to boost the electron transfer and form a stable *p*-conjugation system. Thereby the doping and porous structure of g-C₃N₄ through a postfunctionalization method may be feasible to achieve superior photocatalytic activity.

Herein, we report a facile strategy to synthesize porous B-doped g-C₃N₄ nanosheets by postfunctionalization with PEI and Ph₄BNa. It is interesting to see that the resulted samples achieve boron doping and show porous structure simultaneously, while induce enhanced photocatalytic activity in photocatalytic hydrogen generation. As a result, an impressive H₂ production rate as high as 4280 μ mol/(h·g) is achieved, which is 10.2 folds higher than that of pristine g-C₃N₄.

1 Experimental

1.1 Materials

Urea (CH₄N₂O), Tetraphenylboron sodium (Ph₄BNa, (C₆H₅)₄BNa), and triethanolamine (C₆H₁₅NO₃) were purchased from Sinopharm Chemical Reagent Beijing Co. Ltd. Polyethyleneimine

 $((CH_2CH_2NH)_n, M.W. 70000, 99\%)$ were purchased from Alfa Aesar (China) Chemicals Co. Ltd. All chemicals were analytical grade and used as received without further purification.

1.2 Preparation of g-C₃N₄

The g-C₃N₄ powders were synthesized via the thermal polycondensation of urea according to the procedures described in literature^[22]. In a typical synthesis, urea (10.0 g) sealed in a crucible was heated at a heating rate of 10 °C /min in a muffle furnace from room temperature (RT) to 550 °C and kept at this temperature for 4 h. The resulting yellow product was collected for further use.

1.3 Preparation of boron-doped g-C₃N₄

2 g g-C₃N₄ powders were immersed in 200 mL of DI water. Subsequent ultrasonication and vigorous stirring were applied to the prepared solutions for 0.5 and 12 h, respectively. After that, 0.2 g PEI was added to the above mixture. After continuously stirring for 1 h, suspension was separated through centrifugation at 10000 r/min, and the supernatants were poured out. The precursor solution of unmodified $g-C_3N_4$ was prepared by diluting the reserved solid into 200 mL of DI water. Then, 20 mg Ph₄BNa solution (1 mg/mL solution in DI water) was dropwise added to the above precursor solution of 20 mL while applying stirring for 1 h. The precursor precipitate was obtained by filtering and dried at 60 °C overnight for further synthesis. Boron-doped g-C₃N₄ was prepared by calcining the precursor precipitate, which was heated to 500 °C at a rate of 5 °C/min and maintained at 500 °C for 4 h in air. The B-doped g-C₃N₄ samples were denoted as BPCNX, in which X was the Ph₄BNa solution of 20, 50, 70, and 110 mg, respectively. For comparison purposes, another B-doped $g-C_3N_4$ (denoted as BCN) was also prepared by mixing the urea and Ph₄BNa directly without Ph₄BNa added according to the above procedures.

1.4 Characterization of B-doped g-C₃N₄

The samples were characterized using X-ray diffraction (Shimadzu XRD-7000 with Cu $K\alpha$ radiation, $\lambda = 0.15418$ nm), scanning electron microscopy (Hitachi S-8020U) and transmission electron microscopy (JEOL JEM-2010, operated at 200 kV). Nitrogen adsorption-desorption isotherms and pore-size distributions were measured on a Micromeritics ASAP 2020 at -196 °C. Specific surface areas and mean pore sizes were calculated according to the BET and Barrett-Joyner-Halenda (BJH) methods, respectively. A Shimadzu UV-

3100 PLUS spectrophotometer was used to measure the UV-visible (UV-Vis) absorption spectra of the samples. X-ray photoelectron spectroscopic (XPS) analysis was performed on a PHI Quantera SXM X-ray photoelectron spectrometer using the Al $K\alpha$ radiation. The photoluminescence (PL) spectra of g-C₃N₄ and B-doped g-C₃N₄ were recorded using the Edinburgh Instruments Xe900 equipped with a xenon (Xe) lamp with an excitation wavelength of 380 nm. The FT-IR spectrum was collected using a FT-IR-8400S spectrometer. Electrochemical and photoelectrochemical activities of samples were recorded on a CHI 660B electrochemical workstation in three-electrode quartz cells. A platinum electrode was used as the counter electrode, a saturated calomel electrode (SCE) was used as the reference electrodes, and g-C₃N₄ and B-doped g-C₃N₄ electrodes on FTO served as the working electrodes in 0.1 mol/L Na₂SO₄ aqueous solution. A 300 W Xe lamp (CHF-XW-300W) with a 420 nm cutoff filter was employed as the visible light source. The amperometric *i-t* curves of the samples were measured at 0 V with light on and light off. The electrochemical impedance spectra (EIS) were determined at 0 V. A sinusoidal ac signal of 5 mV was applied to the electrode in the frequency range of 1-100 kHz.

1.5 Photocatalytic experiments

Photocatalytic water splitting was carried out in a top-irradiation vessel connected to a LabSolar H₂ photocatalytic hydrogen evolution system (Perfectlight, Beijing) including a 300 W Xe lamp with a 420 nm cutoff filter. In a typical reaction, 50 mg of photocatalyst powder was dispersed in 100 mL aqueous solution containing 10% triethanolamine scavenger and 1% (respect to Pt, acting as co-catalysts) H₂PtCl₆. 6H₂O. The temperature of the reaction solution was carefully maintained at (5 ± 1) °C during the whole experiment. The reactor was then sealed and evacuated 40 min to remove air before irradiated. The amount of evolved H₂ was analyzed by gas chromatography (Agilent, 7890A GC system) equipped with a thermal conductive detector (TCD) and a 0.5 nm molecular sieve column, using high-purity nitrogen as the carrier gas.

2 Results and discussion

The crystal structure of $g-C_3N_4$ and B-doped $g-C_3N_4$ photocatalysts were examined by XRD, as illustrated in Figure 1(a). The diffractogram of the samples featured two distinct (100) and (002)

reflections at $2\theta = 13.08^{\circ}$ and 27.58°, corresponding to an in-plane structural packing motif and the long-range interplanar stacking of the conjugated aromatic system, respectively^[23]. The XRD patterns of BPCN samples were similar to that of pristine $g-C_3N_4$, however, as for the doped samples, the intensity of the (002) crystal plane reduced substantially with the increase of boron amount, and the (002) diffraction of BPCN-110 (the heaviest doped $g-C_3N_4$) was significantly weakened, which can be attributed to the presence of excessive B atoms doping in the graphitic structure. This can be an evidence for the successful doping in g-C₃N₄. The morphology of the as-prepared g-C₃N₄ samples was examined by field-emission SEM (FE-SEM). The SEM image of BPCN-70 in Figure 1(b) showed a clearly wrinkle 2D structure. As shown in inset of Figure 1(b), the nanosheet was composed of uniform distributed C, N, B and O elements, indicating that B elements were homogeneously doped in g-C₃N₄. The nanostructures of g-C₃N₄ and BPCN-70 were characterized by transmission electron microscopy (TEM). BPCN-70 showed thin nanosheet structure with mesoporous structure (Figure 1(c)), indicating that the postfunctionalization process developed in this work can achieve the delamination and formation of mesopore. This thin nanosheet and uniform mesoporous structure was highly desirable for photocatalytic reaction because the thin layer reduced the distance of charge transfer and pore structure can provide the photocatalytic activity sites. To further confirm the mesopore structure of B-doped g-C₃N₄, textural properties of g-C₃N₄ and BPCN-70 were investigated by the nitrogen adsorption-desorption measurement. The obtained absorption isotherms and Barrett-Joyner-Halenda (BJH) pore-size distributions were shown in Figure 1(d). The adsorption-desorption isotherm of these two samples have been classified as type IV with a H3 hysteresis loop, which was characteristic of mesoporous materials with slit-shaped pores^[24]. The surface area (S_{BET}) and pore volume of BPCN-70 was calculated as to be 125 m^2/g and 0.35 cm^3/g , whereas that of $g-C_3N_4$ was 101 m²/g and 0.49 cm³/g. It was reasonable to state that the increased BET surface area was mainly derived from post-functional treatment, which would provide more active sites for photocatalytic hydrogen evolution. The pore-size distribution curves calculated using the BJH model clearly showed that g-C₃N₄ material had two pore-size families located at 3.9 and 30-50 nm, while the

primary pore size of BPCN-70 was centered at 13 nm (inset of Figure 1(d)), which agreed well with that estimated from TEM images. This pore structure was attributed to the formation of uniform mesoporous structure. These results indicated that the post-

functionalization of g-C₃N₄ yielded a large surface area and mesoporosity. B-doped g-C₃N₄ prepared by other doping amount presented regular S_{BET} values, which were 116, 110, 99 m²/g for the BPCN-20, BPCN-50 and BPCN-110 samples, respectively.



Figure 1 (a) XRD patterns of g-C₃N₄ and BPCN samples; (b) SEM and (c) TEM image of BPCN-70 nanosheet, inset in (b): the element mapping of C, N, B and O for BPCN-70, respectively; (d) Nitrogen adsorption-desorption isotherms, inset: pore size distributions of g-C₃N₄ and BPCN-70

Figure 2(a) depicted the absorbance spectra of g- C_3N_4 and B-doped g- C_3N_4 , which all featured a notable absorption extending from UV part to the visible light region. In comparison with that of $g-C_3N_4$, there was a little red shift of the absorption in BCN and BPCN samples, which may due to the existence of Boron species. The band gaps of g-C₃N₄, BCN-70 and BPCN-70 materials were 2.92, 2.87 and 2.72 eV. BPCN-70 exhibited typical VB characteristics of g-C₃N₄, with the VBM energy at about 1.51 eV versus RHE^[25]. Since the band gap of BPCN-70 was 2.72 eV from the optical measurement, the CBM can be estimated at about -1.21 eV. As a result, the material gap can be narrowed and the ability of BPCN-70 to harvest visible light can be improved with this post-functional modification. The improved photoabsorption property demonstrated that more electrons generated and participated in the photocatalytic reaction, which can result in enhanced photocatalytic activity.

The chemical structure of g-C₃N₄ nanosheets was

confirmed by the FT-IR spectra shown in Figure 2(b). The broad peaks between 3500 and 3000 cm^{-1} originating from the N-H stretches can be clearly observed, suggesting the partial hydrogenation of some nitrogen atoms in the nanosheets. Characteristic bands of aromatic CN heterocycles are at 1200 to 1600 cm⁻¹ and 802 cm⁻¹ is characteristic of s-triazine derivatives^[24,26]. Specifically, the vibrations of the Brelated group (N-B-N) were observed at 1370 cm⁻¹ in the doped CN matrix, despite overlapped of the band with that of the C-N vibrations^[27]. Clearly, the FT-IR spectrum of B doped g-C₃N₄ nanosheets was similar to that of original g-C₃N₄, indicating that the B doped g-C₃N₄ nanosheets keep the same chemical structure as Furthermore, their parent $g-C_3N_4$. the thermogravimetric analysis (TGA-DSC) has been performed to analyze the stability of BPCN-70, as shown in Figure 2(c). The first stage with 8.75% mass loss corresponds to the release of water molecules adsorbed and oxygen-containing groups^[28]. When the temperature was higher than 500 °C, the structure of BPCN-70 began to break up and BPCN-70 sample can be completely decomposed into small molecules such as C_2N_2 and $NH_3^{[29]}$. In addition, the temperature for the maximum endothermic peak of BPCN-70 was at 684 °C, resulting in a strong thermal stability.



Figure 2 (a) UV-Visible diffuse reflectance spectra of g-C₃N₄, BCN-70 and BPCN-70 samples; inset: transformed Kubelka-Munk function versus the energy spectra of g-C₃N₄, BCN-70 and BPCN-70 samples; (b) FT-IR spectra of g-C₃N₄, BCN-70 and BPCN-70 samples; (c) TG spectra of BPCN-70

To further determine the incorporation of B atoms and their oxidation state in the CN framework, The XPS analysis was performed and the results are shown in Figure 3. The XPS survey spectrum in Figure 3(a) showed that the sample contains only C, N and O species without other impurities, while peak of B cannot be noted in the survey spectrum due to the lower intensity compared to that of N specie. A clear B 1s XPS peak with a binding energy $(E_{\rm B})$ of 191.88 eV was observed for the BPCN-70 sample (Figure 3 (b)), corresponding to N-B-N coordination in the unit of heptazine^[30]. This indicated that some of the B atoms were introduced into the carbon sites of CN matrix. The C 1s spectrum (Figure 3(c)) of the BPCN-70 sample displayed two peaks at $E_{\rm B} = 285.48$ and 288.88 eV, which was attributed to contaminated carbon and surface carbon, and the C-(N)3 groups of g- C_3N_4 , respectively^[31]. Figure 3(d) showed the N 1s XPS spectrum of the BPCN-70. This spectrum has been fitted to four peaks, which are attributed to the sp^2 hybridized nitrogen (C-N=C) species at $E_{\rm B}$ = 399.08 eV, the N-(C)3 species at $E_{\rm B} = 400.08$ eV, the quaternary N bonded to three C atoms in the aromatic cycles at $E_{\rm B} = 401.02$ eV and charging effects at $E_{\rm B} =$ 404.58 eV^[32]. According to the XPS and FT-IR analyses, it was concluded that the boron element was indeed doped in g-C₃N₄ nanosheets.

The photocatalytic H₂ production activity of B doped g-C₃N₄ was evaluated and compared with g- C_3N_4 . As shown from H₂ production rates over B doped $g-C_3N_4$ samples with different B contents (Figure 4(a)), the photocatalytic H₂ production rate increased dramatically with B doping. It reached maximum 4280 μ mol /(h·g) over BPCN-70 that showed 10.2-fold improvement in comparison with pure g-C₃N₄ (420 μ mol/(h·g)). The H₂-evolution rate achieved in this work was one of the best for boron doped g-C₃N₄ (Table 1). Moreover, the long-term ability of BPCN-70 was further evaluated (Figure 4(b)). The recycle kinetics curves demonstrated a slight deactivation with time in the first five runs. When an appropriate amount of triethanolamine (TEOA) was added to the reaction solution, BPCN-70 can maintain sustainably stable photocatalytic H₂ production rate over 5 cycles accumulated 20 h, which indicate that it has the good durability and reusability. The photocatalytic performance of g-C₃N₄, BPCN-70 and BCN-70 with no cocatalyst was shown in Figure 5(c). $g-C_3N_4$ presented a H₂-generation rate of 70 μ mol/(h·g), which is much lower than that with Pt. Similarity, with no Pt as cocatalyst, the BPCN-70 sample still presented the fastest H₂-production rate of 610 μ mol/(h·g), which approaches 8.7 times that of pure g-C₃N₄. Additionally, this value is also higher than that of the BCN-70 $(220 \,\mu mol/(h \cdot g)).$



Figure 3 ((a), (b), (c) and (d)) XPS spectra of survey, B 1s, C 1s and N 1s, respectively, for BPCN-70 sample



Figure 4 (a) Photocatalytic H₂-generation activity over the typical samples with 1% Pt as cocatalyst; (b) cyclic running kinetics curves of H₂ production over BPCN-70; (c) photocatalytic H₂-generation activity over the typical samples with no cocatalyst

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Sample	H ₂ -evolution rate/(μ mol·h ⁻¹ ·g ⁻¹)	Condition	Ref.
1	1880	1% Pt, 300 W Xe lamp with a 400 nm cutoff filter	[33]
2	3880	1% Pt, A Xe lamp (350 nm $< \lambda < 780$ nm)	[34]
3	1439	1% Pt, 300 W Xe lamp	[35]
4	704.5	3% Pt, 300 W Xe lamp with 420 nm cutoff filter	[36]
5	278	3% Pt, 300 W Xe lamp with 420 nm cutoff filter	[27]
6	4280	1% Pt, 300 W Xe lamp with 420 nm cutoff filter	this work





Figure 5 (a) transient photocurrent density versus time; (b) photoluminescence spectra; (c) EIS Nyquist plots; (d) time-resolved transient PL spectra of typical samples

To further investigate the underlying mechanism for improved photocatalytic activity, the effect of doping species on the charge carrier migration and separation behaviors were investigated by transient photocurrent responses, EIS and PL spectra. Seen from Figure 5(a), BPCN-70 produced the continually stable transient photocurrent response during the three on/off intermittent irradiation cycles over 210 s under the visible light. It can be seen that the photocurrent of BPCN-1 sample was about 2.4 and 1.2 times higher than that of $g-C_3N_4$ and BCN-70, respectively, suggesting that B doping species by post-modification can efficiently promote the carrier separation and the enhanced photocatalytic activity can be ascribed to the significantly improved photogenerated charge separation efficiency. PL spectra of B doped g-C₃N₄

and g-C₃N₄ samples were recorded at an excitation wavelength of 380 nm to determine the transfer and separation efficiency of photogenerated electron-hole pairs^[37]. As shown in Figure 5(b), the main emission band was centered at about 460 nm for pure g-C₃N₄, which was due to the recombination process of selftrapped excitation^[38]. The emission peak positions of BCN-70 and BPCN-70 samples were similar to those of g-C₃N₄. However, the emission intensity of BPCN-70 composite sample was lower than that of pristine g- C_3N_4 and BCN-70. This result clearly indicates that the recombination of photogenerated charge carriers was inhibited and B-doping into the g-C₃N₄ lattice contributed to the separation of photoinduced charge carriers, which was in agreement with the photocatalytic H₂ evolution results. Moreover, the EIS

spectra in Figure 5(c) presented that the lesser arc radius of Nyquist plot was produced on BPCN-70 electrode compared with g-C₃N₄, proving that the charge carrier recombination was suppressed and interfacial charge transfer ability was promoted effectively, leading to a higher photocatalytic hydrogen production performance. The time-resolved transient PL spectra in Figure 5(d) were further carried out to investigate charge separation behaviors. The fitted fluorescence lifetime increased from 4.64 ns of g-C₃N₄ to 5.76 ns of BPCN-70, which further verified that the doped B significantly prolonged the PL lifetime and hindered the annihilation of charge carriers. Thus, more electrons can participate in the photocatalytic H₂evoultion reaction, which may enhance the performance of BPCN.

Based on the above results, the post-

functionalization process can achieve B-doping into the structure and induce formation of mesoporous structures. The synergistic effect of element doping and mesostructured controlling promotes charge carrier separation and transfer, leading to an enhanced photocatalytic activity.

To gain insight into the stability of BPCN-70, XRD and FT-IR are carried out on BPCN-70 after cycle runs. It can be noted that no obvious change in material structures was observed through XRD and FT-IR examinations (Figure 6(a) and 6(b)) for the sample before and after reaction. This indicates that the decrease in the activity after the first run is mainly due to the decreased concentration of triethanolamine. Based on the above discussion, it is reasonable to conclude that the structure of $g-C_3N_4$ with postfunctional modification is quite stable.



Figure 6 (a) XRD patterns of BPCN-70; (b) FT-IR spectra of BPCN-70 before and after hydrogen evolution reaction

3 Conclusions

In summary, a feasible post-functionalization method has been applied to modify $g-C_3N_4$ to achieve B doping and mesoporous structure, which exhibited enhanced photocatalytic hydrogen production performance. The superior photocatalytic activity and stability were ascribed to the greatly improved charge migration and separation ability. The optimized BPCN-

70 exited optimal H₂ production rate of 4280 μ mol/(h·g) that was 10.2-fold improved in comparison with pristine g-C₃N₄ (420.0 μ mol/(h·g)). Meanwhile, the cycle running experiments indicated that BPCN-70 had good stability and reusability. The findings demonstrate a useful strategy to achieve element doping and structure controlling simultaneously for enhanced photocatalytic performance.

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基于后功能化工艺修饰类石墨相氮化碳及其光催化产氢性能研究

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摘 要:本研究设计了一种后功能化工艺方法修饰类石墨相氮化炭材料。通过此工艺成功得到了硼掺杂的介孔氮化炭材料,该材料比表面积高达 125 m²/g,这为提升光催化分解水性能奠定了基础。利用 X 射线衍射、X 射线光电子能谱,荧光光谱和紫外-可见光谱对材料进行了全面的表征。基于 X 射线光电子能谱分析,发现通过后功能化处理硼原子成功掺杂进入氮化炭的晶格中;通过吸收光谱分析得知,硼掺杂的介孔氮化炭材料增强了在可见光区的光吸收;通过荧光光谱分析得知,相比原始氮化炭材料,硼掺杂后的介孔氮化炭材料有着更低的荧光强度,意味着光生电子和空穴的分离得到了提升。对材料进行光催化分解水测试,后功能化处理得到的硼掺杂介孔氮化炭材料的产氢速率是原始氮化炭材料的10.2 倍。此结论对后续利用后功能化工艺修饰材料提升材料性能具有一定的借鉴意义。

关键词:硼掺杂;介孔结构;后功能化;光催化产氢

中图分类号: O644 文献标识码: A